Study the Gamma ray attenuation parameters of some phosphate glass containing various amount of BCD

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ABSTRACT

By-Pass Cement Dust, as an industrial cement waste, was used here to prepare some gamma-ray shielding glasses aiming to protect people and environment from hazard radiations some phosphate glasses containing different amounts of by-pass cement dust have been prepared by the melt quenching method. The selected molecular composition was \((100-x)\%\ P_2\text{O}_5 - (x)\%\ \text{By-Pass Cement Dust (where } 30 \leq x \leq 60)\]. The obtained experimental density and molar volume values were inspected and were then compared with those obtained empirically for the close packed structure of the corresponding compounds. These comparisons evidenced the short-range order and randomness character of the studied samples. Also X-ray diffraction analyses confirm the amorphous nature of all the prepared glasses.

The suitability of such glasses to act as gamma-ray shielding materials was also examined and a correlation between the chemical composition (By-Pass Cement Dust content) and gamma-ray attenuation behavior was established.

The electric and dielectric properties were thoroughly investigated.
The appearance of maxima and minima in the total conductivity by pass cement dust concentration dependence can be attributed to the mixed alkali–alkaline earth effect ($K_2O$ & $CaO$).

INTRODUCTION:

Glasses are considered now as interesting solid materials due to their interesting properties and functional applications. Among various types of glasses, phosphate glasses are technologically important materials due to their interesting properties [1-3] but their poor chemical durability limits their diverse uses. It was found that the addition of at least 30 mol % metal oxides improves their chemical durability and act to attenuate $\gamma$-ray [4-6].

From another point of view, by-bass cement dust (BCD), as an industrial waste, represents a dangerous by-product of cement industry and it accumulated with huge amounts in Egypt. It causes various diseases, especially those related to human respiratory system [7]. According to the chemical analysis, such waste consists of various oxides (mainly, $CaO$, $SiO_2$, $Na_2O$, $K_2O$ and $Fe_2O_3$), where all these oxides can be introduced into the glass batches during the process of manufacturing different types of oxide glass [8,9]

It was found also that the use of different radioactive isotopes in various daily life fields is spread now, and it is not easy to protect ourselves from various hazard radiations prevailing through the environment. Knowing that, for different nuclear radiation sources, special shielding materials are required. Since glasses are usually transparent and
can be easy manufactured therefore, many articles have been published dealing with γ-ray attenuation of various glasses as transparent shielding materials [10-13].

However, in this article, it will be tried to prepare some phosphate glasses with different additives of BCD, as high as possible, in order to obtain chemically and mechanically stable glasses to consume the waste accumulation as well as to obtain low cost glasses. These glasses will be thoroughly investigated from gamma-ray shielding parameters point of view in order to investigate whether they can be used as transparent γ-ray shields or as capsules for the radio-active wastes before interment underground or they may need some additives of any heavy metal oxides.

Experimental Details:

BCD was firstly analyzed by using X-ray fluorescence (XRF) and its chemical composition was presented in Table, [1].

Table (1), The chemical composition of the used BCD

<table>
<thead>
<tr>
<th>Constituent</th>
<th>CaO</th>
<th>K₂O</th>
<th>SiO</th>
<th>Fe₂O₃</th>
<th>Al₂O₃</th>
<th>MgO</th>
<th>Na₂O</th>
<th>* LOI</th>
</tr>
</thead>
<tbody>
<tr>
<td>Amount %</td>
<td>60.63</td>
<td>10.74</td>
<td>6.96</td>
<td>3.60</td>
<td>1.54</td>
<td>0.03</td>
<td>16.59</td>
<td></td>
</tr>
</tbody>
</table>

* LOI is loss of ignition

The selected glasses were prepared by weighting suitable amounts from ammonium di-hydrogen ortho-phosphate with different additives of the supplied BCD, so that when melted, they supply glasses having the following
composition [(70-x) P₂O₅ - x BCD, where 30 ≤ X≤ 60 in steps of five mol% [11].

The glass batches were ground and mixed well in an agate mortar and they were then transferred into porcelain crucibles inserted into an electric muffle furnace at RT. The temperature of the furnace was raised up to 1100°C during 1 hr and kept at this temperature for 2 hrs. Melts were stirred several times during melting and were then poured between two copper plates in air. The obtained glass samples -just after solidifying- were transferred to the annealing furnace at 450°C for 2 hrs and the furnace was then turned off and left to cool to RT with a cooling rate of about 1.5°C/min.

The experimental density (ρ_exp) was measured by applying Archimedes principle, using carbon tetra-chloride (CCl₄) as an immersion liquid. In such principle, a sample was weighted in air (M_a) and in CCl₄ (M_l), and then (ρ_exp) can be calculated by using equation (1) [12],

\[
ρ_{exp} = \left[ \frac{M_a}{M_a - M_l} \right] ρ_l
\]  

(1)

The experimental molar volume values (V_m)exp were then obtained by using equation (2) [12],

\[
(V_m)_{exp} = \frac{M_m}{ρ_{exp}}
\]  

(2)

Where M_m is the main molecular weight in (g/mol) of a glass sample.

The empirical density values (ρ_emp) were also calculated by using the following relation [13],
\[ \rho_{\text{emp}} = \sum \rho_i X_i \quad (3) \]

where \( \rho_i \) are the densities of the oxides forming a glass sample and \( X_i \) are the mole fractions of each oxide.

The empirical molar volume \( (V_m)_{\text{emp}} \) values were then obtained by using equation (2), but with replacing \( \rho_{\text{exp}} \) by \( \rho_{\text{emp}} \) [13].

XRD patterns were obtained by using Rigaku-RINT 2100, outfitted with Cu \( k_\alpha \) radiation of \( \lambda = 0.1541 \) nm and the operating current and voltage were 300 mA and 50 KV, respectively.

Gamma-ray attenuation parameters (total mass attenuation coefficient \( (\mu_L/\rho)_m \), half value layer (HVL) and mean free path (MFP) of a glass sample were then calculated by applying WIN-XCOM program, based on the mixture rule [13].

Different amount of BaO as heavy metal oxide were then introduced at the expense of \( \text{P}_2\text{O}_5 \) in the samples \([(50-x) \text{P}_2\text{O}_5 - x\text{BaO} - 50 \text{BCD}, \text{where} \ 0 \leq X \leq 20 \text{ in steps of five mol\%}. \ The \ linear \ attenuation \ coefficients \ of \ the \ studied \ glass \ system \ were \ measured \ in \ narrow \ beam \ transmission \ geometry. \ The \ experimental \ setup \ of \ such \ geometry \ was \ exhibited \ in \ Ref. \ [14]. \n
For electrical measurements, the obtained solid glasses were polished from both sides in order to obtain optically flat disk shape samples of 8 mm diameter and 1 mm thickness. The disks were then coated from both sides with an air-drying silver paste to achieve good electrical contact. The measurements were carried out by using a computerized Stanford LCR bridge
model SR 720 at four fixed frequencies \([0.12, 1, 10, 100 \text{ kHz}]\). All measurements were performed in the temperature range from RT up to 250°C.

**RESULTS AND DISCUSSION.**

1- **Confirmation of the amorphous nature**

All the obtained solid samples were examined visually where they appeared transparent and free of inclusions and air bubbles, that is they appeared in good and homogeneous glassy phase. In spite of this, they were examined by using XRD analysis and the obtained patterns are exhibited in Fig. (1) Where all patterns are found free of any sharp crystalline peak [15]

![XRD patterns](image)

Fig. (1) The obtained XRD patterns of all glasses
Both the obtained density values ($\rho_{\text{exp}}$ & $\rho_{\text{emp}}$) are listed in Table (2), for comparison as a function of BCD content, where it is seen that both $\rho_{\text{exp}}$ & $\rho_{\text{emp}}$ increased gradually with the gradual increase of BCD.

Table (2) The obtained $\rho_{\text{exp}}$ and $\rho_{\text{emp}}$ values as a function of BCD content.

<table>
<thead>
<tr>
<th>BCD mol%</th>
<th>30%</th>
<th>35%</th>
<th>40%</th>
<th>45%</th>
<th>50%</th>
<th>55%</th>
<th>60%</th>
</tr>
</thead>
<tbody>
<tr>
<td>$\rho_{\text{exp}}$ (gm/cm³)</td>
<td>1.92</td>
<td>2.35</td>
<td>2.47</td>
<td>2.62</td>
<td>2.70</td>
<td>2.85</td>
<td>3.16</td>
</tr>
<tr>
<td>$\rho_{\text{emp}}$ (gm/cm³)</td>
<td>2.70</td>
<td>2.75</td>
<td>2.80</td>
<td>2.85</td>
<td>2.90</td>
<td>2.95</td>
<td>3.20</td>
</tr>
</tbody>
</table>

Since the molar volume is directly related to the internal spatial structure of materials, it is suitable to exhibit also the change of the molar volume as a function of BCD of the studied glasses. Both molar volume values ($V_{\text{m emp}}$ & $V_{\text{m exp}}$) are exhibited in Table (3), as a function of BCD.

Table (3) The calculated $V_{\text{exp}}$ and $V_{\text{emp}}$ values as a function of BCD content.

<table>
<thead>
<tr>
<th>BCD mol%</th>
<th>30%</th>
<th>35%</th>
<th>40%</th>
<th>45%</th>
<th>50%</th>
<th>55%</th>
<th>60%</th>
</tr>
</thead>
<tbody>
<tr>
<td>$V_{\text{exp}}$ (gm/cm³)</td>
<td>59.95</td>
<td>47.36</td>
<td>43.16</td>
<td>39.06</td>
<td>36.26</td>
<td>32.86</td>
<td>28.28</td>
</tr>
<tr>
<td>$V_{\text{emp}}$ (gm/cm³)</td>
<td>42.82</td>
<td>40.43</td>
<td>38.12</td>
<td>35.90</td>
<td>33.75</td>
<td>31.68</td>
<td>27.87</td>
</tr>
</tbody>
</table>

It is observed that both molar volume values show approximately gradual decrease. The observed variations of both density and molar volume values may be due to the introduced BCD which contains various positive cations. These cations fill mostly the network vacancies, and in turn decrease the internal free volume.

Accordingly, the density is logically increased while the molar volume decreased. In addition, it can be stated that, the higher empirical density and the lower empirical molar volume values than these obtained experimental...
can be taken as confirmation for the amorphous nature and the short-range order character of the studied samples [16].

**2- Gamma-Ray Attenuation parameters:**

Until now, the majority of nuclear radiation shields in nuclear facilities are formed mainly of layers of different types of concrete with various compositions and densities as well as layers of lead sheets. But considerable variations of water content in concrete induces high uncertainty factor in calculating their attenuation coefficients due to the continuous variations of their densities, while lead is characterized by its toxicity. Moreover, both concert and lead are opaque to visible light [17]. But materials that have to be used for shielding must be of stable density and composition as well as they preferably to be transparent. In this regard, glasses are promising materials, since they are usually transparent and easy manufactured.

Berger and Hubbell [18] at 1987 have developed a program called X-COM program for calculating the total mass absorption coefficients or the photon interaction cross-sections for any element, compounds or mixtures at various photon energies from 1 keV to 100 GeV. Recently, X-COM was transformed to the Windows platform by Gerward et al. [19] at 2001 where it is named Win-XCom. With the development of such program, it becomes easy to calculate the total mass attenuation coefficient for different shielding materials. However, it is of interest to check the ability of the studied glasses to act as transparent shielding materials and to investigate whether their
compositions are suitable for this purpose or they may need to introduce some heavy metal oxides into their networks.

On the other hand, the linear attenuation coefficient of the studied glasses have been measured experimentally by applying the narrow beam transmission geometry, where if an incident $\gamma$-ray photon of intensity ($I_0$) have passed through a sample of thickness ($x$) the exit photon intensity ($I$) can be calculated by using the following equations

$$\mu_L = \ln\left(\frac{I_0}{I}\right) \quad (4)$$

The experimental mass attenuation coefficient can be calculated by using both the measured $\mu_L$ & $\rho_{exp}$ as follows equation,

$$\mu_m = \ln\left(\frac{I_0}{I}\right) / \rho_x \quad (5)$$

On the other hand the value of the total mass attenuation coefficient of the studied glasses can be calculated by applying Win-X-COM program (based on the mixture rule) by using equation (6) [20,21].

$$\left(\frac{\mu}{\rho}\right)_m = \sum w_i \left(\frac{\mu}{\rho}\right)_{mi} \quad (6)$$

Where $(\mu/\rho)_m$ is the total mass attenuation coefficient of a sample, $(\mu/\rho)_{mi}$ is the total mass attenuation coefficient of the composing elements of such sample and $w_i$ is the fractional weights of each element in the studied sample.
Theoretical values for the total mass attenuation coefficient can be found in the tables prepared by Hubbell and Seltzer [20].

The linear attenuation coefficients ($\mu_L$) can be then calculated and the obtained values are correlated to the half value layer (HVL) according to equation (7),

$$HVL = 0.693/ \mu_L$$  \hfill (7)

Where HVL is the thickness of the material that decreases the intensity of the incident photon to its half value.

The Mean free path (MFP) of a gamma-ray photon ($\lambda$), can be also calculated by using equation (8),

$$\lambda_{(MFP)} = 1/\mu_L$$  \hfill (8)

Where ($\lambda$) is the mean distance that a photon can travel between any two successive interactions of the photon with matter.

Fig. (2) shows the variation of the calculated $\mu_m$ values for the studied glasses at different low $\gamma$-ray energies as a function of BCD content. It is found that $\mu_m$ show generally, slight linear increase with the increase of the weight fraction of BCD at the expense of $P_2O_5$. This may be due to the gradual increase in the density of the investigated glass samples. Such increase in density act to increase the photon interaction probability at these energies which indicates better shielding properties. The onset plotted between $\mu_m$
and BCD content at γ–ray energy at 356 Kev show obviously such variations.

Fig. 2. The variation of $\mu_m$ as a function of BCD content at different low γ–ray energies

The numerical values of the calculated $\mu_m$ at different energies are summarized in Table (4).

The linear attenuation coefficient ($\mu_L$) represents the fraction of photons removed from a monoenergetic beam per unit thickness of a material and it is expressed in units of cm$^{-1}$. Fig. (3) Shows the variation of linear attenuation coefficient as a function of BCD content, at different low gamma-ray energies. It appeared that, $\mu_L$ increases with the increase of BCD, which may be due also to the gradual increase of density.
Table 4. The variation of the Mass attenuation coefficient as a function of BCD content at relatively low γ-ray energies

<table>
<thead>
<tr>
<th>BCD mol%</th>
<th>Density gm/cm$^3$</th>
<th>$\mu_m$ (cm$^2$/gm) X (10$^{-2}$)</th>
<th>356 Kev</th>
<th>662 Kev</th>
<th>1173 Kev</th>
<th>1332 Kev</th>
</tr>
</thead>
<tbody>
<tr>
<td>30</td>
<td>1.93</td>
<td>9.851</td>
<td>7.44</td>
<td>5.64</td>
<td>5.29</td>
<td></td>
</tr>
<tr>
<td>35</td>
<td>2.35</td>
<td>9.86</td>
<td>7.47</td>
<td>5.67</td>
<td>5.32</td>
<td></td>
</tr>
<tr>
<td>40</td>
<td>2.48</td>
<td>9.87</td>
<td>7.50</td>
<td>5.69</td>
<td>5.34</td>
<td></td>
</tr>
<tr>
<td>45</td>
<td>2.62</td>
<td>9.88</td>
<td>7.53</td>
<td>5.72</td>
<td>5.37</td>
<td></td>
</tr>
<tr>
<td>50</td>
<td>2.71</td>
<td>9.89</td>
<td>7.56</td>
<td>5.76</td>
<td>5.40</td>
<td></td>
</tr>
<tr>
<td>55</td>
<td>2.85</td>
<td>9.9</td>
<td>7.59</td>
<td>5.79</td>
<td>5.43</td>
<td></td>
</tr>
<tr>
<td>60</td>
<td>3.16</td>
<td>9.91</td>
<td>7.64</td>
<td>5.82</td>
<td>5.45</td>
<td></td>
</tr>
</tbody>
</table>

Figure 3. The variation of the $\mu_L$ values as a function of BCD content at different γ-ray energies

Fig. (4) shows the behavior of HVL for the studied glasses as a function of BCD, at various low gamma-ray energies. This figure indicates that the HVL decreases with the increase of the weight fractions of BCD. This may be due to the gradual increase of the linear attenuation coefficient and the
densities of the studied glass samples. Also, the onset in this figure shows obviously such variation.

![Figure 4](image_url)

**Figure 4.** The variation of the HVL as a function of BCD content at relatively low γ–ray energies.

The numerical values of the calculated HVL at different low γ–ray energies are summarized in Table (5).

**Table 5.** The calculated HVL as a function of BCD, at relatively low gamma-ray energies

<table>
<thead>
<tr>
<th>BCD (mol%)</th>
<th>Density (gm/cm³)</th>
<th>HVL (cm)</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td></td>
<td>356 Kev</td>
</tr>
<tr>
<td>30</td>
<td>1.92</td>
<td>6.93</td>
</tr>
<tr>
<td>35</td>
<td>2.35</td>
<td>6.92</td>
</tr>
<tr>
<td>40</td>
<td>2.47</td>
<td>6.91</td>
</tr>
<tr>
<td>45</td>
<td>2.62</td>
<td>6.89</td>
</tr>
<tr>
<td>50</td>
<td>2.70</td>
<td>6.89</td>
</tr>
<tr>
<td>55</td>
<td>2.85</td>
<td>6.88</td>
</tr>
<tr>
<td>60</td>
<td>3.16</td>
<td>6.87</td>
</tr>
</tbody>
</table>
The MFP is the average distance traveled by a moving photon between two successive impacts. Such value can be used to describe the effectiveness of the studied shields. Fig. (5), shows the behavior of the MFP as a function of BCD content, at various low gamma-ray energies. This figure indicates that MFP decreases with the increase of the weight fractions of BCD, which may be due to the gradual increase of the molar volume of the studied glass samples, and hence the gradual increase of $\mu_L$ [19].

![Figure 5. The variation of MFP, as a function of BCD content](image)

It is suitable to compare the obtained values (exp. & Theo.) of the linear attenuations coefficients for the studied glasses

It can be observed from Table. (6) that the experimental linear attenuation coefficient exhibits approximately gradual increase as BCD was increased, but it shows approximately gradual decrease as the $\gamma$-ray energy was increased.
Table. (6) The experimental & theoretical linear attenuation coefficient

<table>
<thead>
<tr>
<th>BCD%</th>
<th>30%</th>
<th>35%</th>
<th>40%</th>
<th>45%</th>
<th>50%</th>
<th>55%</th>
<th>60%</th>
</tr>
</thead>
<tbody>
<tr>
<td>$\mu_{\text{exp}}$</td>
<td>0.0762</td>
<td>0.0942</td>
<td>0.0942</td>
<td>0.1122</td>
<td>0.12</td>
<td>0.1302</td>
<td>0.1821</td>
</tr>
<tr>
<td>$\mu_{\text{theo}}$</td>
<td>0.065</td>
<td>0.085</td>
<td>0.105</td>
<td>0.126</td>
<td>0.144</td>
<td>0.16</td>
<td>0.17</td>
</tr>
</tbody>
</table>

It can be observed from Fig. (6) that the comparison between the exp. & theo. linear attenuation coefficient of the studied of glasses exhibits approximately gradual increase as BCD was increased, it appeared also that both variations are very near to each other and the observed difference may be due to the used industrial waste.

![Figure 6. Comparison between the exp. & Theo. linear attenuation of glasses.](image-url)
The total mass attenuation coefficient of the samples containing BaO

Noticing that, the studied glasses contain no any heavy metal cations (only phosphorous and BCD constituting cations), and it is supposed that, if some heavy metal cations are introduced into their network, this may improve their gamma-ray attenuation parameters. However, these parameters will be checked again when different amounts of BaO are introduced into their networks. The mass attenuation coefficients as a function of γ-ray energies for different BaO concentrations are represented in Table. (6).

Table 6. The variation of the Mass attenuation coefficient as a function of BaO content at relatively low γ-ray energies.

<table>
<thead>
<tr>
<th>BaO mol%</th>
<th>356 Kev</th>
<th>662 Kev</th>
<th>1173 Kev</th>
<th>1332 Kev</th>
</tr>
</thead>
<tbody>
<tr>
<td>0</td>
<td>10.1</td>
<td>7.68</td>
<td>5.83</td>
<td>5.47</td>
</tr>
<tr>
<td>5</td>
<td>10.3</td>
<td>7.69</td>
<td>5.81</td>
<td>5.44</td>
</tr>
<tr>
<td>10</td>
<td>10.5</td>
<td>7.70</td>
<td>5.78</td>
<td>5.41</td>
</tr>
<tr>
<td>15</td>
<td>10.7</td>
<td>7.71</td>
<td>5.75</td>
<td>5.39</td>
</tr>
<tr>
<td>20</td>
<td>10.9</td>
<td>7.72</td>
<td>5.72</td>
<td>5.37</td>
</tr>
</tbody>
</table>

It can be observed from Fig. (7) that the mass attenuation coefficient exhibits approximately gradual increase as BaO was increased, but it shows approximately gradual decrease as the γ-ray energy was increased. This behavior shows good confirmation that the introduced BaO act to increase the mass attenuation coefficient. It appeared also that the studied glass samples exhibit high shielding efficiency at low γ-ray energies and their efficiency decreased as the γ-ray energy increased.
Figure 7. The variation of $\mu m$ of glasses containing different concentration of BaO.

It can be observed from Table. (7) that the comparison between the exp. & Theo. linear attenuation coefficient of glasses containing heavy metals exhibits also approximately gradual increase as BaO was increased.

Table (7) Shows the obtained values of the Theo. & exp. linear attenuation coefficient of glasses containing BaO

<table>
<thead>
<tr>
<th>BaO %</th>
<th>5</th>
<th>10</th>
<th>15</th>
<th>20</th>
</tr>
</thead>
<tbody>
<tr>
<td>Theo. $\mu$</td>
<td>.1122</td>
<td>.1211</td>
<td>.1302</td>
<td>.1425</td>
</tr>
<tr>
<td>exp. $\mu$</td>
<td>.1224</td>
<td>.1323</td>
<td>.1432</td>
<td>.1546</td>
</tr>
</tbody>
</table>

Fig. (8) show a comparison between the exp. linear attenuation coefficient of glasses containing BaO and glass without BaO. And both variations exhibits approximately gradual increase as BaO was increased. it can be observed also that the linear attenuation coefficient of the glasses containing BaO are higher than those containing no BaO.
Figure 8. Comparison between exp. linear attenuation of the glasses containing different amounts of BaO content.

Generally, these results indicated that the studied glasses are good attenuators for gamma photons and they represent promising gamma-ray shielding materials due to their high mass attenuation coefficient and their low half value layer, especially when some BaO has introduced into their glass networks.

3. conductivity

It is well know fact that, the interaction of gamma-ray with matter can induces some electrical charges. These charged must be readily released from the shielding material directly. Therefore checking the semiconductor properties of such glass is of interest. The dc conductivity values were then calculated from the fitting of the experimentally obtained $\sigma_T$ data with equa-
tion (9) and the obtained data are then plotted in Fig. (9) as a function of BCD content.

\[ \sigma_{dc} = \sigma_0 \exp \left( -\frac{\Delta E_a}{kT} \right) \]  

(9)

where \( \sigma_0 \) is the pre-exponential factor, \( \Delta E_a \) is the activation energy, \( k \) is Boltzmann constant and \( T \) is the absolute temperature.

![Figure 10. The variation of \( \sigma_{dc} \) as a function of BCD](image)

It is seen that the variation of \( \sigma_{dc} \) is a non-linear behavior and a maxima and a minima are exhibited at 35 and 50 mol\% BCD, corresponding to CaO concentrations of 21 and 33 mol\% respectively and the amounts of K\(_2\)O are 3.3 and 6.05 mol\% respectively. It can be supposed that the observed non-linear behavior may be due to the mixed alkali-alkaline earth effect [20, 23].

The dc electrical activation energy (\( \Delta E \)) can be then calculated form the slopes of the obtained straight lines of the studied sample at relatively high temperatures (where the dc conductivity is dominant) according to Arhenius equation (equation 8), The variation of the activation energy with BCD is exhibited in Fig. (10), where this variation shows approximately the reverse
behavior of the dc conductivity and such trend was expected to be logic [24, 25].

![Graph showing the variation of ∆E values as a function of BCD](image)

**Figure 10.** The variation of ∆E values as a function of BCD

**CONCLUSION:**

According to the obtained results and the proposed discussion in this study, the following conclusions can be drown:

1- Some phosphate glasses with different additives of BCD (industrial iron wastes) as high as 60 mol%, can be prepared, where these glasses are environmentally friends.

2- The visual examination showed pure glassy phase of all the prepared samples, and the XRD analysis confirm their amorphous nature as well as their randomness structure.
3- The densities of the studied glasses are found to increase gradually as BCD was gradually increased.

4- The studied glasses represent good attenuators for γ –photons and promising gamma-ray shielding material due to their high mass attenuation coefficient and their low half value layer (HVL) only at low gamma-ray energies.

5- The introduced of some heavy metal cation improve all the gamma-ray shielding parameters.

6- all the studied glasses show semi conduction properties and they can release any charge that can be formed on these shielding glasses.

REFERENCES


